Temperature stability of field induced anisotropy in soft ferromagnetic Fe,Co-based amorphous and nanocomposite ribbons

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The temperature stability of field induced uniaxial anisotropy (KU) was investigated by thermomagnetic treatments of (Co1−xFe)x89Zr7B4 amorphous ribbons after field annealing below and above the crystallization temperature. We conclude: (1) Field annealing treatments are necessary to properly investigate the temperature stability of KU, (2) KU of field crystallized alloys exhibit improved temperature stability relative to alloys remaining amorphous after field annealing, and (3) larger KU is obtained for field crystallization treatments as compared to zero-field crystallization followed by field reannealing. Field crystallization may be required for elevated temperature applications when field induced anisotropy is critical for performance. © 2009 American Institute of Physics. [DOI: 10.1063/1.3068547]

Magnetic field annealing1,2 is used to tailor magnetic properties of soft magnetic materials by inducing a uniaxial magnetic anisotropy energy, KU. The induced easy axis can be controlled by applying a saturating magnetic field (i.e., field annealing). However, induced anisotropy due to “self-field annealing” also develops in the absence of an external field resulting in a spatially varying distribution of induced anisotropies that stabilize the remanent domain structure of the material. Field annealing treatments are particularly important for soft ferromagnetic Fe,Co-based amorphous and nanocomposite alloys. The effective magnetocrystalline anisotropy is greatly reduced by exchange coupling of regions with rapid spatial variations (over ~10−10−10−9 m) of local easy axis orientations. KU due to induced magnetic anisotropies can dominate the effective magnetocrystalline anisotropy because of greater spatial coherency.4–6

Toroidal tape wound cores of amorphous or nanocomposite ribbons can be annealed in a longitudinal or a transverse saturating field. Large KU is often undesirable for zero or longitudinally field annealed cores as it can dominate the effective magnetic anisotropy and result in large hysteretic losses due to domain wall pinning.4 However, for transverse field annealed cores large KU can be desirable resulting in higher saturation fields, lower permeabilities, and improved high frequency response.7,8

KU in Fe,Co-based soft magnetic amorphous and amorphous/nanocrystalline “nanocomposite” alloy systems has been reported and discussed in a number of previous works.5–12 Co-rich nanocomposites tend to exhibit relatively large values of KU as compared to the corresponding Fe-rich compositions. (Fe,Co)–Zr–B is one of the most widely studied alloy systems due to high saturation magnetizations and Curie temperatures. We have previously reported the compositional dependence of KU for transverse field annealed (Co1−xFe)x89Zr7B4 and (Co1−xFe)x88Zr7B4Cu9,10 in this work, we discuss the temperature stability of KU using subsequent thermomagnetic treatments of selected ribbons from our previous reports. The ribbons were initially field annealed at temperatures above or below the primary crystallization temperature, Ts, ~ 450 °C. Emphasis is placed on the (Co0.972Fe0.028)89Zr7B4 composition for which a large value of KU (~20000–25000 ergs/cm3) was observed after field crystallization to produce BCC, FCC, and HCP nanocrystals surrounded by an intergranular amorphous phase.9,10,13 Results are also discussed for alloys with varying Fe, Co content and the (Co0.88Fe0.12)70Cu1.9,B2,7SG8 composition investigated by other authors.11,12

Amorphous ribbons were synthesized by arc-melting and single roller wheel melt spinning. Toroidal wound cores were annealed in flowing N2 for 1 h with a 2 T transverse field. The cores were then reannealed with or without a longitudinal applied saturating field of H > 50 Oe under flowing N2. Dynamic B-H loops at f = 3 kHz were measured using ac permeametry to estimate the anisotropy field, HK, by extrapolating the low-field linear portion of the B-H loop to saturation. The room temperature saturation magnetization, Ms, was measured by vibrating sample magnetometry. HK and Ms were used to estimate the value of KU using the expression KU = (HKMs)/2. Phase identification and microstructural investigations are discussed in detail elsewhere.9,10,13

A dramatic difference in the temperature stability of KU was observed for zero field and longitudinal field reannealing of initially transverse field annealed ribbons. This is illustrated in Fig. 1 for a (Co0.972Fe0.028)89Zr7B4 alloy after a Tanneal = 350 °C transverse field annealing treatment (field annealed amorphous). Figure 1(a) shows measured values of HK at room temperature after reannealing the sample at increasing temperatures for 1 h. Because Ms of the as-cast and

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crystallized ribbons are almost identical for this composition.

The trend in $H_K$ of Fig. 1 also illustrates the trend in $K_U$. For zero-field reannealing treatments, the initial values of $H_K$ are 18 Oe and $K_U$ is 8400 ergs/cm$^3$ decrease slightly with increasing reannealing temperatures. An increase occurs upon reannealing at temperatures sufficiently high for crystallization to occur due to the larger values of $H_K$ and $K_U$ for field crystallized alloys of this composition. This observation demonstrates that the zero-field reannealing treatments are effectively a “self-transverse field anneal” due to the high Curie temperature of the as-cast amorphous ribbon and a transverse remanent domain structure established by the initial transverse field anneal.

Measured $B$-$H$ loops after the initial transverse field annealing and after zero-field reannealing up to 450 °C to result in crystallization are presented in Fig. 1(b). The $B$-$H$ loops are less linear for such “self-transverse field crystallized” alloys as compared to alloys crystallized in a transverse, saturating external field [e.g., Fig. 2(a)] presumably due to the more complex induced anisotropy distribution (e.g., in the vicinity of domain walls). For comparison, the $B$-$H$ loops are also presented for an initially transverse field annealed amorphous ribbon before and after longitudinal field reannealing at $T_{\text{anneal}} = 250$ °C in Fig. 1(c). The transverse field induced anisotropy is completely eliminated even at this low annealing temperature demonstrating that “self-field annealing” effects must be avoided by applying a saturating longitudinal or rotating magnetic field to properly investigate the temperature stability of $K_U$ through reannealing treatments. Therefore we subsequently focus only on results obtained for longitudinal field reannealing treatments.

Figure 2 shows the results of longitudinal field reannealing treatments for the high $K_U$ compositions (Co$_{0.975}$Fe$_{0.025}$Zr$_{0.025}$B$_4$ [Fig. 2(a)] and (Co$_{0.88}$Fe$_{0.12}$Zr$_{0.025}$B$_4$ [Fig. 2(b)]) after transverse field crystallization at $T_{\text{anneal}} = 540$ °C. For the field crystallized alloys, $M_S$ does not change significantly during reannealing and so $H_K$ again represents the trend in $K_U$. Both field crystallized alloys exhibit improved temperature stability as compared to the field annealed amorphous alloy of Fig. 1 for which measurable values of $H_K$ were eliminated after a longitudinal field reannealing treatment at temperatures as low as $T_{\text{anneal}} = 250$ °C. A difference in temperature stability can potentially be explained by a higher atomic mobility in the as-cast amorphous ribbons. For the (Co$_{0.88}$Fe$_{0.12}$Zr$_{0.025}$B$_4$ alloy, the $B$-$H$ loops exhibit a “kink” after longitudinal field reannealing so the estimates of $H_K$ are a slight, but consistent, overestimate of the field at which the core saturates.

A correlation between $H_K$ and coercivity ($H_C$) is not observed for the (Co$_{0.975}$Fe$_{0.025}$Zr$_{0.025}$B$_4$ alloy [Fig. 2(e)] but a rough correlation is observed for the (Co$_{0.88}$Fe$_{0.12}$Zr$_{0.025}$B$_4$ alloy [Fig. 2(d)]. A correlation between $H_K$ and static values of $H_C$ would be reasonable in the framework of the extended random anisotropy model where $K_T$ dominates the overall effective anisotropy. However, we cannot address this issue here as the measured values of $H_C$ are dynamic and there is some ambiguity in using this model to explain the coercivities of transverse field annealed cores where magnetization occurs primarily through rotation.

In Fig. 3(a), $H_K$ is presented as a function of longitudinal field reannealing temperature for several different alloys. Figure 3(b) demonstrates the total reduction in $H_K$ after reannealing treatments for 1 h while Fig. 3(c) presents the fractional reduction in $H_K$. These results illustrate that, with the exception of the x=0.15 alloy, all of the field crystallized ribbons exhibit enhanced temperature stability of $K_U$ as compared to the field annealed amorphous ribbon of Fig. 1. The
relatively low temperature stability demonstrated by the
(Co_{0.83}Fe_{0.15})_{89}Zr_{2.6}B_{9}Cu_{1} (x=0.15) field crystallized alloy is
not understood at this time.

The temperature stability of $K_U$ can provide information
about the potential mechanisms responsible for the field
induced anisotropy. Further investigation of the temperature
stability of $K_U$ is worthwhile in an attempt to clarify the
origin of field induced anisotropy in the complex nanocom-
posites. A comparison between the relative stability of $K_U$ for
alloys in which different mechanisms of field induced aniso-
tropy are thought to be dominant could be particularly en-
lightening.

A final set of experiments demonstrate the importance of
applying a magnetic field during crystallization to generate
the largest $K_U$. B-H loops were compared for ribbons crys-
tallized in a transverse field ($T_{\text{anneal}}=540$ °C for 1 h) with
those obtained for ribbons crystallized without a field and
then subsequently reannealed in a transverse field (same tem-
perature and time). In Fig. 4(a), B-H loops measured for
the zero-field crystallized ($\text{Co}_{0.975}\text{Fe}_{0.025})_{89}Zr_{2.6}B_{9}$ ribbon are
presented and compared to the transverse field crystallized core.
No measurable $H_K$ is found after transverse field anneal-
ing of the zero-field crystallized ribbon, but there is a large $H_K$
=$43$ Oe ($K_U$=$22,000$ ergs/cm$^3$) for the transverse field
crystallized ribbon. Similar results have been obtained for
other Co:Fe ratios (for example, $x=0.00, x=0.50$, and $x$
=1.00) and in all cases transverse field crystallization treat-
ments are more effective than transverse field reannealing of
zero-field crystallized ribbons. For the high $K_U$ ($\text{Co}_{0.88}$
$\text{Fe}_{0.12})_{79.4}\text{Nb}_{2.6}\text{Si}_{9}\text{B}_{9}$ alloy a measurable $H_K$=$22$ Oe ($K_U$
$\sim9000$ ergs/cm$^3$) was induced by transverse field reanneal-
ing treatment but it was lower than the $H_K$=$43$ Oe ($K_U$
=$17,500$ ergs/cm$^3$) obtained after transverse field crystal-
lization. We suggest that these observations are likely due to a
higher atomic mobility during the early stages of primary
crystallization.

Temperature stability of field induced anisotropy, $K_U$, in
amorphous and nanocrystalline/amorphous nanocomposite
alloys was investigated. We conclude that (1) field annealing
 treatments are necessary to investigate the temperature sta-
bility of $K_U$, (2) $K_U$ of field crystallized alloys typically ex-
hibit higher temperature stability than alloys remaining
amorphous after field annealing, and (3) larger $K_U$ is ob-
tained for alloys crystallized in a field rather than reannealed
in a field after zero-field crystallization. Field crystallization
may be required for elevated temperature applications when
field induced anisotropy is critical for performance. Exper-
iments examining the temperature stability of $K_U$ for field
crystallized alloys deserve further investigation as they may
help provide insight into the mechanistic origin of $K_U$ in
these complex systems.

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